



# Preparation and electrochemical performance of Na<sup>+</sup> and Co<sup>2+</sup> co-doped Li<sub>0.9</sub>Na<sub>0.1</sub>Mn<sub>1-x</sub>Co<sub>x</sub>PO<sub>4</sub>/C cathode material for Li-ion battery

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## Abstract

In this work, we have successfully in-situ synthesized Na<sup>+</sup> and Co<sup>2+</sup> co-doped Li<sub>0.9</sub>Na<sub>0.1</sub>Mn<sub>1-x</sub>Co<sub>x</sub>PO<sub>4</sub>/C nanoparticles on the surface of Li<sub>2.7</sub>Na<sub>0.3</sub>PO<sub>4</sub> self-sacrificing template by the co-precipitation process combined with the hydrothermal method. The crystal lattice structure, crystal appearance and electrochemical parameters are characterized by X-ray diffractometer (XRD), scanning electron microscopy (SEM), galvanostatic charge and discharge test, cyclic voltammetry (CV) and electrochemical impedance spectroscopy (EIS). SEM analysis indicates that Li<sub>0.9</sub>Na<sub>0.1</sub>Mn<sub>0.9</sub>Co<sub>0.1</sub>PO<sub>4</sub>/C composite shows uniform porous structure and nanosized grain particles. The electrochemical measurements show that the double ions co-doping routine plays a vital influence on the rate capability and electrochemical lithium storage property of LiMnPO<sub>4</sub> material. The initial discharge specific capacity of Li<sub>0.9</sub>Na<sub>0.1</sub>Mn<sub>0.9</sub>Co<sub>0.1</sub>PO<sub>4</sub>/C reaches 164.3 mAh/g (0.05 C) and 148.0 mAh/g (1 C), respectively. The excellent rate capability is attributed to the synergetic doping effect of Na<sup>+</sup> and Co<sup>2+</sup> on improving the Li-ion diffusion rate and broadening the Li-ion diffusion channels.

**Keywords** Lithium ion battery · LiMnPO<sub>4</sub> · Na<sup>+</sup> and Co<sup>2+</sup> co-doping effect · Electrochemical performance

## Introduction

As new energy vehicles and electrochemical energy storage devices develop dramatically, the growing demands have been proposed on developing high energy/power densities lithium-ion power battery to satisfy people's production and living needs [1–5]. The cathode material is a critical ingredient of lithium-ion battery (LIBs); among the olivine phosphate family, LiMnPO<sub>4</sub> has been regarded as a significant cathode over commercial LiFePO<sub>4</sub> due to the superiority of discharge voltage platform, specific energy density (700 Wh/kg), superior safety and low price [6–9], which could also match the existing electrolytic liquid system. However, the high Li-ion diffusion activation energy and the one-dimensional channel

lead to sluggish electronic/ionic conductivity and inferior electrochemical performance of LiMnPO<sub>4</sub> [10, 11]. The crystal structure transformation between LiMnPO<sub>4</sub>/MnPO<sub>4</sub> caused by the Jahn-Teller effect would induce the collapse of lattice structure and the inferior cycle performance. To date, a good deal of modification methods [12–26] have been adopted to ameliorate electrochemical lithium storage property of LiMnPO<sub>4</sub>/C. In general, the cation doping of LiMnPO<sub>4</sub> cathode with similar ionic radius to Mn<sup>2+</sup> is an effective means to break through the barrier, the strategy of mutual cations doping at Li/Mn site could widen the one-dimensional Li-ion diffusion path, maintain the crystal structure stability and accelerate the Li-ion diffusion velocity [27, 28]. The selection of dopant is also a critical factor, the Co<sup>2+</sup> (0.74 Å) has the similar ion radius to Mn<sup>2+</sup> (0.80 Å), which is the main factor for choosing Co<sup>2+</sup> as a dopant to inhibit lattice volume expansion and ameliorate the Jahn-Teller distortion of Mn<sup>3+</sup> [29, 30]. Wang et al. [31] has synthesized the gradient Co-doped LiMn<sub>0.98</sub>Co<sub>0.02</sub>PO<sub>4</sub> by the secondary solvothermal method, the capacity retention of LiMn<sub>0.98</sub>Co<sub>0.02</sub>PO<sub>4</sub> is remarkably improved with the percentage of 87% after 380 cycles, the improvement is mainly due to the preferential growth along [010]-oriented LiMnPO<sub>4</sub> with more Li<sup>+</sup> migration channels and reaction sites. The Na-doped Li<sub>0.9</sub>Na<sub>0.1</sub>MnPO<sub>4</sub>/C material

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[32] and the  $\text{Na}^+$  and  $\text{Ni}^{2+}$  co-doped  $\text{Li}_{0.9}\text{Na}_{0.1}\text{Mn}_{1-x}\text{Ni}_x\text{PO}_4/\text{C}$  cathode materials [23] have been studied in my research group, which show excellent discharge capacity and cycle stability. With the deepening of research, various methods have emerged to prepare double ions co-doped polyanionic-based  $\text{LiMnPO}_4/\text{C}$  cathode materials [33–38]. Huang et al. [33] have prepared  $\text{Fe}^{2+}$  and  $\text{Ti}^{4+}$  co-doped  $\text{LiMnPO}_4/\text{C}$  cathodes by a solid-state reaction route, the synergistic effect greatly enhances the electrochemical performance. Ramar and Balaya [34] have adopted a ball mill-assisted soft template method to synthesize the isovalent co-doped  $\text{LiMn}_{0.9}\text{Fe}_{0.05}\text{Mg}_{0.05}\text{PO}_4/\text{C}$  sample, which shows higher lithium storage capacity (159 mAh/g, 0.1 C) and better cycling stability compared to  $\text{LiMnPO}_4$  doped with either  $\text{Fe}^{2+}$  or  $\text{Mg}^{2+}$ . Recently, Li et al. [35] have adopted a solvothermal method to prepare  $\text{Li}_{1-x}\text{Na}_x\text{Mn}_{0.8}\text{Fe}_{0.2}\text{PO}_4/\text{C}$  nanocapsule; after 200 cycles, the capacity retention ratio is nearly 96.65% (0.5 C). Among these, hydrothermal approach has been selected as a high efficiency and good controllability synthesize method to optimize the electrochemical performances. Therefore, the strategy of cobalt ions and sodium ions co-doped at Mn/Li site seems to be a prospective attempt to enhance the electrochemical lithium storage performance of  $\text{LiMnPO}_4$  through affording more lithium ions diffusion channels and reaction sites.

In this paper, we have adopted co-precipitation method to prepare  $\text{Li}_{2.7}\text{Na}_{0.3}\text{PO}_4$  self-sacrificial template; subsequently, the nanosized  $\text{Li}_{0.9}\text{Na}_{0.1}\text{Mn}_{1-x}\text{Co}_x\text{PO}_4/\text{C}$  cathode materials have been in-situ synthesized on the surface of  $\text{Li}_{2.7}\text{Na}_{0.3}\text{PO}_4$  by a facile hydrothermal reaction. The microstructure, grains size and electrochemical characteristics of  $\text{Li}_{0.9}\text{Na}_{0.1}\text{Mn}_{1-x}\text{Co}_x\text{PO}_4/\text{C}$  are systematically researched to evaluate the co-doping effect of  $\text{Na}^+$  and  $\text{Co}^{2+}$  on improving the electrochemical properties of  $\text{LiMnPO}_4/\text{C}$ .

## Experimental

### Preparation

The different cobalt ion doping amounts of  $\text{Li}_{0.9}\text{Na}_{0.1}\text{Mn}_{1-x}\text{Co}_x\text{PO}_4/\text{C}$  cathodes have in-situ synthesized through the hydrothermal method, using as-prepared  $\text{Li}_{2.7}\text{Na}_{0.3}\text{PO}_4$  and  $\text{MnSO}_4 \cdot \text{H}_2\text{O}$  as mainly raw materials,  $\text{Co}(\text{NO}_3)_2 \cdot 6\text{H}_2\text{O}$  as doping material. The  $\text{Li}_{2.7}\text{Na}_{0.3}\text{PO}_4$  precursor has been successfully prepared in my previous work [32]. Other raw materials were analytically pure and commercially available without any treatment. To clarify the influence of cobalt ion doping on the electrochemical parameters of  $\text{Li}_{0.9}\text{Na}_{0.1}\text{Mn}_{1-x}\text{Co}_x\text{PO}_4/\text{C}$  cathode materials, parallel experiments were conducted on maintaining all parameters that remain the same and only changing the contents of cobalt ions.

The synthesis process of  $\text{Li}_{0.9}\text{Na}_{0.1}\text{Mn}_{1-x}\text{Co}_x\text{PO}_4/\text{C}$  was described as follows. Firstly, the  $\text{Li}_{2.7}\text{Na}_{0.3}\text{PO}_4$ ,  $\text{MnSO}_4 \cdot \text{H}_2\text{O}$  and  $\text{Co}(\text{NO}_3)_2 \cdot 6\text{H}_2\text{O}$  were, respectively, resolved in the  $\text{PEG400-H}_2\text{O}$  ( $V_{\text{PEG400}}/V_{\text{H}_2\text{O}}=12\text{ ml}/24\text{ ml}$ ) mixture solvents in a 1:1-x:x mole ratio, then the mixture solvents kept vigorous stirring for about 30 min to obtain the homogeneous solution. The above mixture solution was dissolved in a 50 ml Teflon-lined hydrothermal autoclave and reacted at 180 °C for 12 h. After the hydrothermal reaction, the synthetic substance was gathered after washing, centrifuged with deionized water and alcohol several times, then dried in the oven at 80 °C for 8 h. Finally, the  $\text{Li}_{0.9}\text{Na}_{0.1}\text{Mn}_{1-x}\text{Co}_x\text{PO}_4/\text{C}$  hydrothermal product was mixed and grinded with ascorbic acid in a mass fraction of 20 wt.% to form uniformly distributed ascorbic acid coated precursor; after calcination at 550 °C for 3 h in the flowing nitrogen gas atmosphere, the target  $\text{Li}_{0.9}\text{Na}_{0.1}\text{Mn}_{1-x}\text{Co}_x\text{PO}_4/\text{C}$  were obtained.

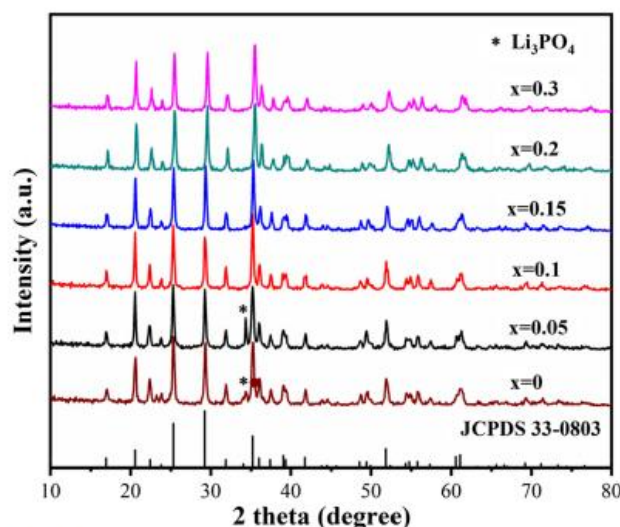
The electrochemical properties of  $\text{Li}_{0.9}\text{Na}_{0.1}\text{Mn}_{1-x}\text{Co}_x\text{PO}_4/\text{C}$  cathodes were measured by CR2032 coin cell, and fabricated in argon filled glove box. The cathode coating slurry was prepared through mixing the cathode material, PVDF binder and acetylene in a 8:1:1 mass ratio in NMP solvent under continuously vigorous stirring for 4 h. The aluminum foil collector was prepared by coating the cathode slurry on the aluminum foil, drying at 80 °C for 10 h in air, and the following 120 °C for 12 h under vacuum condition, aiming to evaporate the NMP and  $\text{H}_2\text{O}$ . After rolling and cutting, the 0.785  $\text{cm}^2$  circular sheet was formed. In the assembly process of CR2032 coin cells, lithium wafer, Celgard 2400 membrane and 1M  $\text{LiPF}_6/\text{EC}+\text{MEC}+\text{DEC}$  (volume ratio=1:1:1) mixture solution was served as the anode, separator and electrolyte in turn.

### Characterization

The microstructures and crystal lattice data were tested by X-ray diffractometer (XRD, Rigaku Smartlab) with  $\text{Cu K}\alpha$  radiation under a sweep rate of 0.04°/s. The appearance of the crystal, grain size, pore structure of the doped materials were investigated by scanning electron microscopy (SEM, ZEISS SUPRA55) and transmission electron microscopy (TEM, JEOL JEM2100F). The concentration of  $\text{Mn}^{2+}$  and  $\text{Co}^{2+}$  resolved in the electrolyte was measured by the inductively coupled plasma-atomic emission spectrometer (ICP-AES, PS-6).

The electrochemical parameters of the CR2032 coin cell was measured by the charge/discharge measurements on the Land CT2001A battery tester within the wide operating voltage range (2.5–4.5 V). The charge transfer kinetics and electrode polarization phenomenon were tested by CV and EIS methods on the Solartron 1260+1287 electrochemical workstation. The EIS test frequency was in the range of  $10^{-1}$ – $10^5$  Hz, and the voltage amplitude was set to 10 mV. The CV test





**Fig. 1** XRD diffraction patterns of  $\text{Li}_{0.9}\text{Na}_{0.1}\text{Mn}_{1-x}\text{Co}_x\text{PO}_4/\text{C}$  ( $x = 0, 0.05, 0.1, 0.15, 0.2, 0.3$ ) samples

voltage range was on the scale of 2.5–4.5 V at a 0.1 mV/s sweep rate. All measurements were tested at ambient temperature.

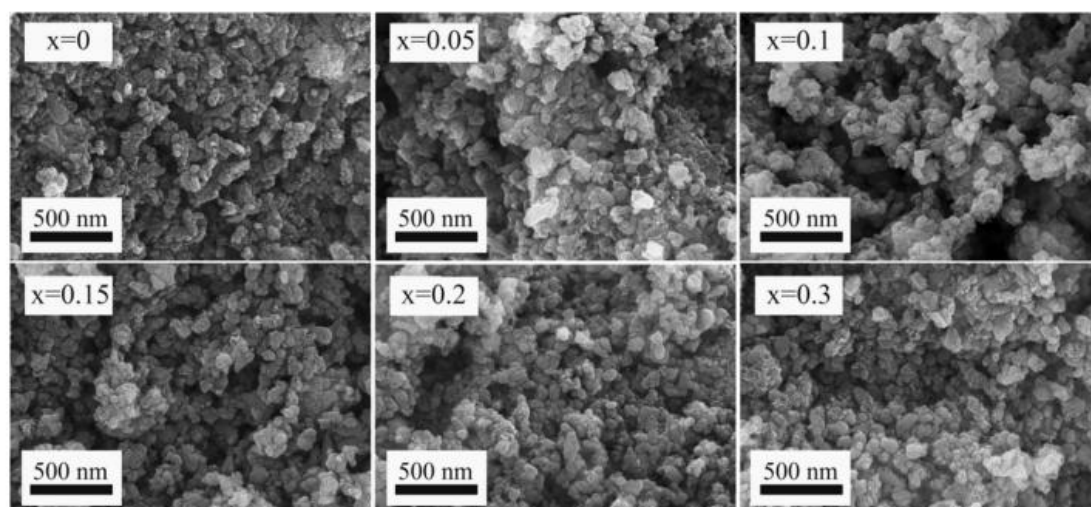
## Results and discussion

The  $\text{Na}^+$  and  $\text{Co}^{2+}$  co-doped  $\text{Li}_{0.9}\text{Na}_{0.1}\text{Mn}_{1-x}\text{Co}_x\text{PO}_4/\text{C}$  samples have been synthesized through the hydrothermal method; in the doped samples, the doping ratio of  $\text{Na}^+$  is fixed at 0.1, and vary the  $\text{Co}^{2+}$  contents ratio ( $x$ ) with 0, 0.05, 0.1, 0.15, 0.2, 0.3. Fig. 1 displays the XRD patterns of  $\text{Li}_{0.9}\text{Na}_{0.1}\text{Mn}_{1-x}\text{Co}_x\text{PO}_4/\text{C}$  samples, and the JCPDS standard card of  $\text{LiMnPO}_4$  (No. 33-0803). The main characteristic diffractive peaks of all doped samples are well indexed with the

orthorhombic system (Pnmb) of  $\text{LiMnPO}_4$ . From the Rietveld lattice parameters of all the  $\text{Li}_{1-x}\text{Na}_x\text{MnPO}_4/\text{C}$  diffraction patterns calculated by High Score Software, we have found that the lattice constant values of Co-doped samples change little with no obvious regularity, which is largely due to the similar ion radius of  $\text{Co}^{2+}$  (0.74 Å) to  $\text{Mn}^{2+}$  (0.80 Å). In the sample of  $x = 0$  and  $x = 0.05$ , an additional diffraction peak is observed at  $34.4^\circ$ , which is associated with  $\text{Li}_3\text{PO}_4$  phase (JCPDS card No. 25-1030), and in accordance with our previous research [32]. There are no evident impurities and carbon diffraction peaks, which means that the pyrolytic carbon exists in amorphous form. This indicates that  $\text{Na}^+$  and  $\text{Co}^{2+}$  have embedded into the crystal lattice of  $\text{LiMnPO}_4$ , and a certain amount of  $\text{Na}^+$  and  $\text{Co}^{2+}$  could not influence the crystal structure of the doped samples. The shape of the diffraction peaks is very narrow and strong mean the high crystallinity of the as-prepared materials.

Figure 2 displays the SEM photographs of different cobalt ion doping amounts of  $\text{Li}_{0.9}\text{Na}_{0.1}\text{Mn}_{1-x}\text{Co}_x\text{PO}_4/\text{C}$  samples. It can be found that the morphologies of all samples present similar microstructures with irregular spherical-like shape primary particles aggregating together. The average diameters of the primary particles are ranging from 40 to 80 nm. When the cobalt doping ratio ( $x$ ) is 0.1, the inner pore structure seems more uniform in distribution, indicating that an appropriate cobalt doping ratio could effectively increase the reactivity between active material and electrolytes across the two-phase interface. The real chemical compositions of  $\text{Li}_{0.9}\text{Na}_{0.1}\text{Mn}_{0.9}\text{Co}_{0.1}\text{PO}_4/\text{C}$  sample are measured by ICP, and the practical molar ratio of  $\text{Li}:\text{Na}:\text{Mn}:\text{Co}:\text{P}$  is 0.907:0.093:0.897:0.103:1.000. The results are nearly the designed stoichiometric ratio of 0.9:0.1:0.9:0.1:1.

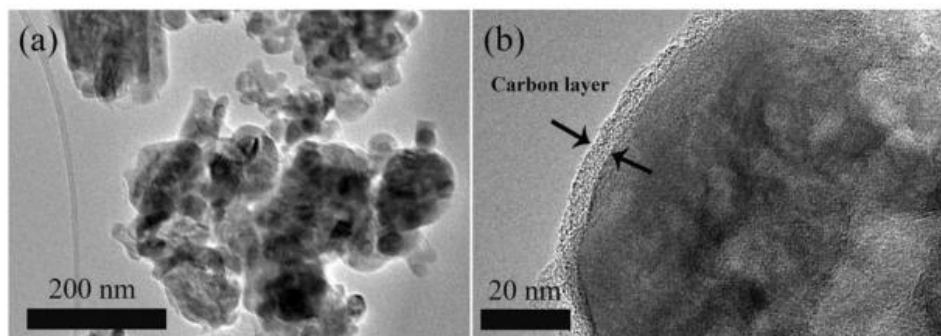
Figure 3 shows the TEM photographs of  $\text{Li}_{0.9}\text{Na}_{0.1}\text{Mn}_{0.9}\text{Co}_{0.1}\text{PO}_4/\text{C}$  sample. Numerous nanosized grains stack tightly together with the average diameters of



**Fig. 2** SEM photographs of as-prepared  $\text{Li}_{0.9}\text{Na}_{0.1}\text{Mn}_{1-x}\text{Co}_x\text{PO}_4/\text{C}$  ( $x = 0, 0.05, 0.1, 0.15, 0.2, 0.3$ ) samples



**Fig. 3** TEM photographs of  $\text{Li}_{0.9}\text{Na}_{0.1}\text{Mn}_{0.9}\text{Co}_{0.1}\text{PO}_4/\text{C}$  sample

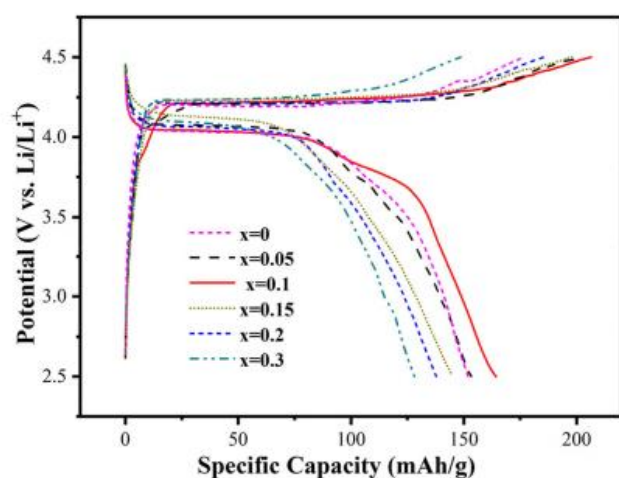


22–58 nm. It can be seen from the high-magnification TEM photograph in Fig. 3(b), the continuous carbon layer is uniformly coated on the surface of the nanoparticles with the thickness of 4–5 nm, which is obtained by the pyrolysis of carbon sources. The uniform conductive carbon layer will facilitate the electronic transfer in the surface layer of the cathode materials.

The galvanostatic charge/discharge measurements of different  $\text{Co}^{2+}$  doping amounts of  $\text{Li}_{0.9}\text{Na}_{0.1}\text{Mn}_{1-x}\text{Co}_x\text{PO}_4/\text{C}$  samples are tested at 0.05 C discharge rate, the charging and discharging curves are displayed in Fig. 4. All the discharging curves exhibit a lank voltage platform nearly 4.05 V (vs.  $\text{Li}/\text{Li}^+$ ), which are attributed to redox reaction between the redox couple of  $\text{Mn}^{3+}/\text{Mn}^{2+}$ . The doping amounts of  $\text{Co}^{2+}$  greatly affect influence the discharge specific capacity of  $\text{Li}_{0.9}\text{Na}_{0.1}\text{MnPO}_4/\text{C}$ . As the  $\text{Co}^{2+}$  doping amounts increase, the initial discharge capacity are 152.0 mAh/g, 153.5 mAh/g, 164.3 mAh/g, 144.6 mAh/g, 137.4 mAh/g and 128.2 mAh/g, respectively. For all samples,  $\text{Li}_{0.9}\text{Na}_{0.1}\text{Mn}_{0.9}\text{Co}_{0.1}\text{PO}_4/\text{C}$  delivers relatively high discharge capacity. The capacity enhancement is attributed to a moderate cobalt ion doping amount that could effectively improve intrinsic conductivity and increase reactive sites during the Li-ion insertion/

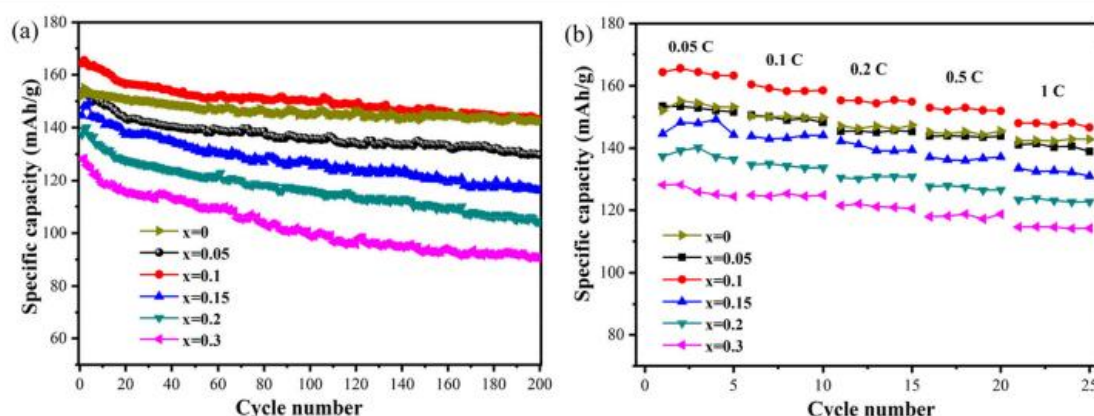
deinsertion process. However, excessive  $\text{Co}^{2+}$  doping contents would cause capacity deterioration. The cycle stability and rate performance are tested to reflect the electrochemical performance of the doped samples. The cycle performance of  $\text{Li}_{0.9}\text{Na}_{0.1}\text{Mn}_{1-x}\text{Co}_x\text{PO}_4/\text{C}$  composites is measured at 0.05 C, and the cyclic curves are shown in Fig. 5(a). After 200 cycles,  $\text{Li}_{0.9}\text{Na}_{0.1}\text{Mn}_{0.9}\text{Co}_{0.1}\text{PO}_4/\text{C}$  and  $\text{Li}_{0.9}\text{Na}_{0.1}\text{MnPO}_4/\text{C}$  decrease to the similar discharge capacity value, but still higher than other cobalt ion doping samples. The conclusions indicate that excessive  $\text{Co}^{2+}$  doping amounts would lead to poor cycle performance. Fig. 5(b) shows the rate capability of the doped samples measured at different rates. Obviously,  $\text{Li}_{0.9}\text{Na}_{0.1}\text{Mn}_{0.9}\text{Co}_{0.1}\text{PO}_4/\text{C}$  electrode shows excellent rate discharge performance, and in different rate, the discharge capacity is 164.3 mAh/g (0.05 C), 160.4 mAh/g (0.1 C), 155.3 mAh/g (0.2 C), 152.9 mAh/g (0.5 C) and 148.0 mAh/g (1 C), respectively. The discharge capacity slightly decreases with the discharge rate increasing. It indicates that a proper amount of  $\text{Co}^{2+}$  doping is beneficial to obtain high rate performance of the cathode material by means of maintaining the structural stability and broadening the lithium ions diffusion channel during the Li-ion insertion/deinsertion process.

The EIS test has been conducted to analyze the reaction kinetics on the interface of two-phase media of  $\text{Li}_{0.9}\text{Na}_{0.1}\text{Mn}_{1-x}\text{Co}_x\text{PO}_4/\text{C}$  composite materials. Fig. 6(a) shows the EIS curves of all the doped samples, the Nyquist curves are composing of a semicircular shape and an oblique line two parts, respectively located in the high-middle frequency range and the low frequency range. The intercept of the semicircular on the axis of reals is corresponding to the complex charge transfer resistance ( $R_{ct}$ ). The slope of the straight line represents the lithium ion diffusion velocity ( $W_0$ ). The value of  $R_{ct}$  is different. The  $\text{Li}_{0.9}\text{Na}_{0.1}\text{Mn}_{0.9}\text{Co}_{0.1}\text{PO}_4/\text{C}$  sample exhibits the smallest  $R_{ct}$  value, indicating that  $\text{Li}_{0.9}\text{Na}_{0.1}\text{Mn}_{0.9}\text{Co}_{0.1}\text{PO}_4/\text{C}$  shows the smallest charge transfer resistance. For further understanding the  $\text{Na}^+$  and  $\text{Co}^{2+}$  co-doping effects on the charge transfer dynamics, we have fitted the EIS results and the linear fitting curves between  $Z'$  and  $\omega^{-1/2}$  are represented inset in Fig. 6(a). According to the equation  $Z' = R_s + R_{ct} + \sigma\omega^{-1/2}$ , we could obtain the Warburg factor ( $\sigma$ ) of all the samples by calculating the slope of the equation. The



**Fig. 4** The first charge/discharge curves of  $\text{Li}_{0.9}\text{Na}_{0.1}\text{Mn}_{1-x}\text{Co}_x\text{PO}_4/\text{C}$  ( $x = 0, 0.05, 0.1, 0.15, 0.2, 0.3$ ) samples at 0.05 C





**Fig. 5** The cycling ability (a) and the rate performance (b) of  $\text{Li}_{0.9}\text{Na}_{0.1}\text{Mn}_{1-x}\text{Co}_x\text{PO}_4/\text{C}$  ( $x = 0, 0.05, 0.1, 0.15, 0.2, 0.3$ ) samples

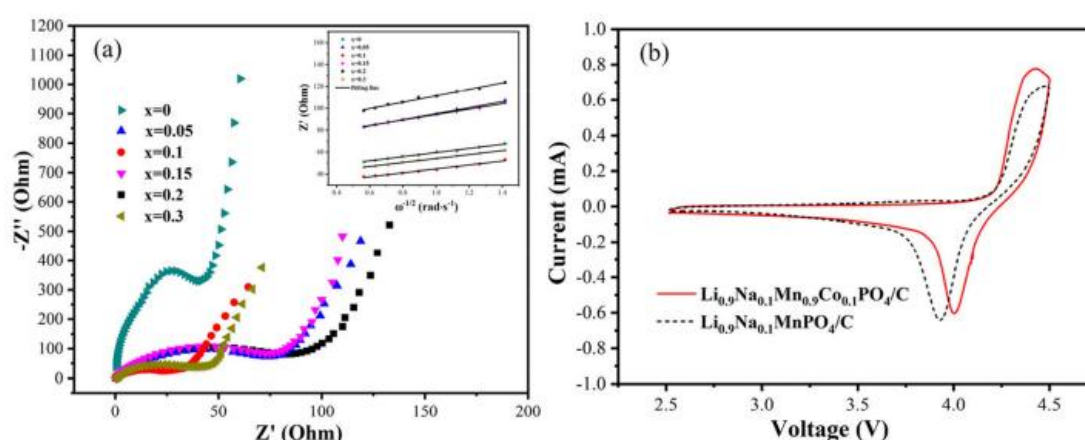
$\text{Li}_{0.9}\text{Na}_{0.1}\text{Mn}_{0.9}\text{Co}_{0.1}\text{PO}_4/\text{C}$  shows the lower  $\sigma$  value, which means that the sample has the higher lithium ion diffusion coefficient ( $D_{\text{Li}^+}$ ). The results show that  $\text{Li}_{0.9}\text{Na}_{0.1}\text{Mn}_{0.9}\text{Co}_{0.1}\text{PO}_4/\text{C}$  has low electrode polarization and excellent reaction kinetics in the Li-ion insertion/deinsertion process. In a word, the synergistic effects of  $\text{Na}^+$  and  $\text{Co}^{2+}$  co-doped are beneficial to Li-ion diffusion in the lattice structure of the cathode materials.

From the CV plots of  $\text{Li}_{0.9}\text{Na}_{0.1}\text{MnPO}_4/\text{C}$  and  $\text{Li}_{0.9}\text{Na}_{0.1}\text{Mn}_{0.9}\text{Co}_{0.1}\text{PO}_4/\text{C}$  samples in Fig. 6(b), a couple of good symmetrical reversible redox peaks have been found, which are assigned to the redox reaction of  $\text{Mn}^{3+}/\text{Mn}^{2+}$  accompanied with the Li-ion insertion/deinsertion process. The redox process is a quasi-reversible reaction. In the cycle process, the redox voltage difference values of  $\text{Li}_{0.9}\text{Na}_{0.1}\text{MnPO}_4/\text{C}$  and  $\text{Li}_{0.9}\text{Na}_{0.1}\text{Mn}_{0.9}\text{Co}_{0.1}\text{PO}_4/\text{C}$  are close to 0.54 and 0.43 V. With the adding of cobalt ions, the potential interval decreases slightly with no serious electrode polarization, illustrating that a proper doping ratio of  $\text{Na}^+$  and  $\text{Co}^{2+}$  is favorable to improve the reaction reversibility and reduce electrode polarization rate of  $\text{LiMnPO}_4/\text{C}$ .

## Conclusions

In this work,  $\text{Na}^+$  and  $\text{Co}^{2+}$  co-doped  $\text{Li}_{0.9}\text{Na}_{0.1}\text{Mn}_{1-x}\text{Co}_x\text{PO}_4/\text{C}$  nanoparticles have been successfully in-situ prepared on the surface of  $\text{Li}_{2.7}\text{Na}_{0.3}\text{PO}_4$  self-sacrificing template through the co-precipitation process combined with the hydrothermal method. A proper doping ratio of  $\text{Co}^{2+}/\text{Na}^+$  is beneficial to reduce the grain size, increase pore size and porosity, and further improve the migration velocity of lithium ion. By comparing with other doped samples,  $\text{Li}_{0.9}\text{Na}_{0.1}\text{Mn}_{0.9}\text{Co}_{0.1}\text{PO}_4/\text{C}$  sample shows the superior high discharge capacity of 164.3 mAh/g (0.05 C) and 148.0 mAh/g (1 C), which exhibit excellent discharge performance induced by the abundant pore structure and the widened Li-ion diffusion channel of the cathode material.

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**Fig. 6** (a) The EIS curves of  $\text{Li}_{0.9}\text{Na}_{0.1}\text{Mn}_{1-x}\text{Co}_x\text{PO}_4/\text{C}$  ( $x = 0, 0.05, 0.1, 0.15, 0.2, 0.3$ ) samples, inset: the linear fitting curves; (b) The CV plots of  $\text{Li}_{0.9}\text{Na}_{0.1}\text{MnPO}_4/\text{C}$  and  $\text{Li}_{0.9}\text{Na}_{0.1}\text{Mn}_{0.9}\text{Co}_{0.1}\text{PO}_4/\text{C}$  samples



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